Fracture Mechanics, Damage and Fatigue Non Linear Fracture Mechanics – HRR Theory

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Fracture Mechanics – NLFM – HRR Theory

Linear Elastic Fracture Mechanics (LEFM)



Asymptotic solution governed by stress intensity factors



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before

- Cracked body: summary
 - Potential energy $\Pi_T = E_{int} Qu$
 - Crack closure integral
 - Energy required to close crack tip

$$\Delta \Pi_T = \int_{\Delta A} \int_{\boldsymbol{u}}^{\boldsymbol{u} + \Delta \boldsymbol{u}} \boldsymbol{t} \cdot [\boldsymbol{u}'] d\boldsymbol{u}' dA$$

- Energy release rate
 - Variation of potential energy in case of crack growth

$$G = -\partial_{\rm A} \left(E_{\rm int} - W_{\rm ext} \right) = -\partial_{A} \Delta \Pi_{T}$$

• In linear elasticity

$$G = -\partial_A \Delta \Pi_T = -\lim_{\Delta A \to 0} \frac{1}{\Delta A} \int_{\Delta A} \frac{1}{2} \mathbf{t}^{\mathbf{0}} \cdot \left[\!\left[\Delta \mathbf{u} \right]\!\right] dA$$



- In linear elasticity & if crack grows straight ahead

$$G = \frac{K_I^2}{E'} + \frac{K_{II}^2}{E'} + \frac{K_{III}^2}{2\mu} \qquad E' = \frac{4E}{(1+\nu)(\kappa+1)} = \begin{cases} E & P\sigma \\ \frac{E}{1-\nu^2} & P\varepsilon \end{cases}$$





- Cracked body: summary
 - J-integral
 - Strain energy flow

$$J = \int_{\Gamma} \left[U\left(\boldsymbol{\varepsilon}\right) \boldsymbol{n}_{x} - \boldsymbol{u}_{,x} \cdot \boldsymbol{T} \right] dl$$

- Exists if an internal potential exists
 - Is path independent if the contour Γ embeds a straight crack tip
 - No assumption on subsequent growth direction
 - Can be extended to plasticity if no unloading (see lecture on cohesive zone)
- If crack grows straight ahead \implies G=J
- In linear elasticity (independently of crack growth direction):

$$J = \frac{K_I^2}{E'} + \frac{K_{II}^2}{E'} + \frac{K_{III}^2}{2\mu} \qquad E' = \frac{4E}{(1+\nu)(\kappa+1)} = \begin{cases} E & P\sigma \\ \frac{E}{1-\nu^2} & P\varepsilon \end{cases}$$





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a

Linear Elastic Fracture Mechanics (LEFM)

- Analytical
 - SIF from full-field solution
 - Limited cases

$$\implies \begin{cases} K_I = \sigma_\infty \sqrt{\pi a} \\ K_{II} = \tau_\infty \sqrt{\pi a} \\ K_{III} = \tau_\infty \sqrt{\pi a} \end{cases}$$



- From energetic consideration
 - Growing straight ahead crack
 - From J-integral
- Numerical (e.g. FEM)

 $\implies \begin{cases} K_I = \beta_I \sigma_\infty \sqrt{\pi a} \\ K_{II} = \beta_{II} \tau_\infty \sqrt{\pi a} \\ K_{III} = \beta_{III} \tau_\infty \sqrt{\pi a} \end{cases}$

- β_i depends on geometry & crack length
- Tabulated solutions (handbooks)

http://ebooks.asmedigitalcollection.asme.org/book.aspx?bookid=230 2021-2022 Fracture Mechanics – NLFM – HRR Theory 5

 $G = -\partial_{A} (E_{int} - W_{ext}) \Longrightarrow G = \frac{K_{I}^{2}}{E'} + \frac{K_{II}^{2}}{E'} + \frac{K_{III}^{2}}{2\mu}$ $J = \frac{K_{I}^{2}}{E'} + \frac{K_{II}^{2}}{E'} + \frac{K_{III}^{2}}{2\mu}$

- Small Scale Yielding assumption
 - LEFM: we have assumed the existence of a K-dominance zone



- This holds if the process zone (in which irreversible process occurs) •
 - Is a small region compared to the specimen size &
 - Is localized at the crack tip
- Validity of this approach?
 - We check the dimensions

$$a, W - a > 2.5 \left(\frac{K_I}{\sigma_p^0} \right)^2$$
 «Process zone size

- Non-linear fracture mechanics
 - Derivation of the LEFM validity criterion
 - Providing solutions when LEFM criterion is not met







Dugdale (1960) & Barenblatt (1962)'s cohesive model



- The plastic zone has actually a different shape in most cases
 - Depends on the stress state, hardening law etc.





- Elasto-plasticity (small deformations)
 - Beyond a threshold the material experiences irreversible deformations
 - Typical behavior at low/room temperature
 - Curves σ - ϵ independent of time
 - At higher temperature creep ...
 - Yield surface

 $f(\boldsymbol{\sigma}) \leq 0 \begin{cases} f < 0: \text{ elastic region} \\ f = 0: \text{ plasticity} \end{cases}$

- Plastic flow
 - Assumption: deformations can be added $d\boldsymbol{\varepsilon} = d\boldsymbol{\varepsilon}^{\mathrm{e}} + d\boldsymbol{\varepsilon}^{\mathrm{p}} \implies d\boldsymbol{\sigma} = \mathcal{H} : d\boldsymbol{\varepsilon}^{\mathrm{e}}$
 - Normal plastic flow $d\varepsilon^{\rm p} = d\lambda \frac{\partial f}{\partial \sigma}$
- Path dependency (incremental equations in d)





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Material behavior: Elasto-plasticity



Material behavior: Elasto-plasticity

- Existence of energy release rate?
 - Free energy

•
$$\Psi(\boldsymbol{\varepsilon}^{e}, \bar{\boldsymbol{\varepsilon}}^{p}) = \frac{1}{2}\boldsymbol{\varepsilon}^{e}: \mathcal{H}: \boldsymbol{\varepsilon}^{e} + h(\bar{\boldsymbol{\varepsilon}}^{p})$$

• $\boldsymbol{\sigma} = \frac{\partial \Psi}{\partial \boldsymbol{\varepsilon}^{e}}$ & $\sigma_{p} = \frac{\partial \Psi}{\partial \bar{\boldsymbol{\varepsilon}}^{p}} = h'$

- But $\boldsymbol{\sigma} = \frac{\partial U}{\partial \boldsymbol{\varepsilon}^e} \neq \frac{\partial U}{\partial \boldsymbol{\varepsilon}}$

 $\partial oldsymbol{arepsilon}^e$

- No unique value of σ for a given ε
- Since $\sigma(\varepsilon^e)$ depends on the history •

 \blacktriangleright neither G nor I exists

- We assume no possible unloading

We recover an energy $U(\varepsilon)$ with $U(\boldsymbol{\varepsilon}) = \frac{1}{2} \boldsymbol{\varepsilon}^{e}(\boldsymbol{\varepsilon}) : \mathcal{H} : \boldsymbol{\varepsilon}^{e}(\boldsymbol{\varepsilon}) + h(\bar{\varepsilon}^{p}(\boldsymbol{\varepsilon}))$

and with (see next slide)

$$\boldsymbol{\sigma} = \frac{\partial U}{\partial \boldsymbol{\varepsilon}}$$











Material behavior: Elasto-plasticity







1.5

0.5

 $0^{\scriptscriptstyle \mathrm{L}}_0$

2

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 σ/σ^0_{p}

- J2-plasticity without unloading
 - Internal energy

•
$$U(\boldsymbol{\varepsilon}) = \frac{1}{2} \boldsymbol{\varepsilon}^{e}(\boldsymbol{\varepsilon}) : \mathcal{H} : \boldsymbol{\varepsilon}^{e}(\boldsymbol{\varepsilon}) + h(\bar{\varepsilon}^{p}(\boldsymbol{\varepsilon}))$$

• With
$$h(\bar{\varepsilon}^p(\boldsymbol{\varepsilon})) = \int_{\bar{\varepsilon}^p} \sigma_p d\bar{\varepsilon}^p$$

- Example: Power law hardening

•
$$\sigma_p\left(\bar{\varepsilon}^{\mathrm{p}}\right) = \sigma_p^0 \left(\frac{\bar{\varepsilon}^p}{\alpha \sigma_p^0/E} + 1\right)^{\frac{1}{n}}$$

- Parameters α & n
- Term $\sigma_p^{0/E}$ represents the elastic deformation before yield

$$\implies U = \frac{\boldsymbol{\varepsilon}^{\mathrm{e}} : \mathcal{H} : \boldsymbol{\varepsilon}^{\mathrm{e}}}{2} + \frac{n\alpha \left(\sigma_{p}^{0}\right)^{2}}{E\left(n+1\right)} \left[\left(\frac{\bar{\varepsilon}^{p}}{\alpha \sigma_{p}^{0}/E} + 1\right)^{\frac{n+1}{n}} - 1 \right]$$





 $\alpha = 2; n = 1$ $\alpha = 4; n = 1$ $\alpha = 2; n = 4$

 $\alpha = 4; n = 4$

 $\alpha = 2; n = 7$

 $\alpha = 4; n = 7$

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 $\varepsilon E / \sigma_{p}^{0}$

Power law

This law can be rewritten in terms of the total deformations



- Parameter *n* •
 - $n \rightarrow \infty$: perfect plasticity
 - $n \rightarrow 1$: "elasticity"
- Doing so requires 2 assumptions
 - There is no unloading
 - As elastic strains are assimilated to plastic strains, the material is incompressible
- Which are satisfied if

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- We are interested only in crack initiation and not in crack propagation
- The stress components remain proportional with the loading
- Elastic deformations are negligible compared to plastic ones



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- Power law (2)
 - Internal potential for J2-elasto plasticity _

•
$$U(\boldsymbol{\varepsilon}) = \frac{1}{2} \boldsymbol{\varepsilon}^{e}(\boldsymbol{\varepsilon}) : \mathcal{H} : \boldsymbol{\varepsilon}^{e}(\boldsymbol{\varepsilon}) + h(\bar{\varepsilon}^{p}(\boldsymbol{\varepsilon}))$$

with $h(\bar{\varepsilon}^{p}(\boldsymbol{\varepsilon})) = \int_{\bar{\varepsilon}^{p}} \sigma_{p} d\bar{\varepsilon}^{p}$

 Here we have a non-linear incompressible response

•
$$\sigma_e = \sigma_p^0 \left(\frac{\bar{\varepsilon}}{\alpha \sigma_p^0 / E}\right)^{\frac{1}{n}}$$

Obeys plastic flow equations •

$$\implies U(\bar{\varepsilon}) = \int_{\bar{\varepsilon}} \sigma_e d\bar{\varepsilon}$$

with
$$n(\varepsilon'(\varepsilon)) = \int_{\overline{\varepsilon}p} b_p u \varepsilon'$$

ere we have a non-linear
compressible response
 $\sigma_e = \sigma_p^0 \left(\frac{\overline{\varepsilon}}{\alpha \sigma_p^0 / E}\right)^{\frac{1}{n}}$
 $\bullet \quad Obeys plastic flow equations$
 $\Longrightarrow U(\overline{\varepsilon}) = \int_{\overline{\varepsilon}} \sigma_e d\overline{\varepsilon}$
 $U(\overline{\varepsilon}) = \frac{n\alpha(\sigma_p^0)^2}{E(n+1)} \left(\frac{\overline{\varepsilon}}{\alpha \frac{\sigma_p^0}{E}}\right)^{\frac{n+1}{n}} = \frac{n\alpha(\sigma_p^0)^2}{E(n+1)} \left(\frac{\sigma_e}{\sigma_p^0}\right)^{n+1}$





• The HRR theory

- From Hutchinson, Rice and Rosengren, 1967-1968
- Assumptions
 - Semi-infinite crack
 - Loading increased monotonically —> only for crack initiation
 - Power law only if plastic deformations are large compared to elastic ones
- J-integral
 - Elasto-plastic model & these assumptions = non-linear elasticity
 - The J-integral can be used
 - It is path independent, so one may choose

$$J = \int_{-\pi}^{\pi} \left(U \boldsymbol{n}_{x} - \boldsymbol{u}_{,x} \cdot \boldsymbol{\sigma} \cdot \boldsymbol{n} \right) r d\theta$$



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• Near fields

- Evaluation of *J*-integral
 - Choice of a circle to evaluate J

$$\implies J = \int_{-\pi}^{\pi} \left(U \boldsymbol{n}_{x} - \boldsymbol{u}_{,x} \cdot \boldsymbol{\sigma} \cdot \boldsymbol{n} \right) r d\theta$$

• Since *J* is path independent, it is the same whatever the value of *r*

$$\implies J = \lim_{r \to 0} \int_{-\pi}^{\pi} \left(U \boldsymbol{n}_x - \boldsymbol{u}_{,x} \cdot \boldsymbol{\sigma} \cdot \boldsymbol{n} \right) r d\theta$$

So the integrant should be
 independent of *r* (at the limit)

$$\implies \lim_{r \to 0} \left(U \boldsymbol{n}_x - \boldsymbol{u}_{,x} \cdot \boldsymbol{\sigma} \cdot \boldsymbol{n} \right) = \frac{h\left(\boldsymbol{\theta} \right)}{r}$$

- The terms involve stress times strain

$$\implies \lim_{r \to 0} \boldsymbol{\sigma} : \boldsymbol{\varepsilon} = \frac{l(\theta)}{r}$$





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- Near fields
 - Because of the path independence of *J*

•
$$\lim_{r \to 0} \boldsymbol{\sigma} : \boldsymbol{\varepsilon} = \frac{l(\theta)}{r}$$

- Generally speaking, the stress tensor can be expanded in a power series

•
$$\boldsymbol{\sigma}(r, \theta) = \sum_{s} r^{s} \hat{\boldsymbol{\sigma}}(\theta, s)$$

where $\hat{oldsymbol{\sigma}}\left(heta,\,s
ight)$ depends only on heta and s

- Let s' be the dominant exponent near the crack tip

$$\implies \boldsymbol{\sigma}\left(r,\,\theta\right)\simeq r^{s'}\hat{\boldsymbol{\sigma}}\left(\theta,\,s'
ight)$$

where $\hat{\boldsymbol{\sigma}}\left(heta,\,s^{\prime}
ight)$ depends only on heta and $s^{\,\prime}$

• Strain field:

$$\bar{\varepsilon} = \frac{\alpha \sigma_p^0}{E} \left(\frac{\sigma_e}{\sigma_p^0} \right)^n \Longrightarrow \quad \boldsymbol{\varepsilon} \simeq \left(r^{s'} \right)^n \hat{\boldsymbol{\varepsilon}} \left(\theta, \, s' \right)$$

- Therefore

•
$$\lim_{r \to 0} \sigma : \varepsilon = \frac{l(\theta)}{r} \implies ns' + s' = -1 \implies s' = \frac{-1}{n+1}$$





- Near fields (2)
 - Near fields are obtained using $s' = -\frac{1}{n+1}$
 - Rewrite stress field

$$\boldsymbol{\sigma}\left(r,\,\theta\right)\simeq r^{s'}\hat{\boldsymbol{\sigma}}\left(\theta,\,s'\right) \implies \boldsymbol{\sigma}\left(r,\,\theta\right)\rightarrow \sigma_{p}^{0}k_{n}\frac{\tilde{\boldsymbol{\sigma}}\left(\theta,\,n\right)}{r^{\frac{1}{n+1}}}$$

- k_n is a plastic stress intensity factor which
 - Allows defining $\tilde{\sigma}(\theta, n)$ independently of the loading amplitude
 - Depends on n
- Limit cases •
 - "Elasticity" ($n \rightarrow 1$):

 \implies LEFM solution in $1/r^{1/2}$ recovered

- Perfect plasticity $(n \rightarrow \infty)$:
- stress remains finite









- Near fields (3)
 - Near fields are obtained using $s' = -\frac{1}{n+1}$
 - Stress field

$$\boldsymbol{\sigma}\left(r,\,\theta\right) \to \sigma_{p}^{0}k_{n}\frac{\tilde{\boldsymbol{\sigma}}\left(\theta,\,n\right)}{r^{\frac{1}{n+1}}}$$

• Strain field

$$\bar{\varepsilon} = \frac{\alpha \sigma_p^0}{E} \left(\frac{\sigma_e}{\sigma_p^0}\right)^n \implies \varepsilon \to \frac{\alpha \sigma_p^0}{E} \frac{k_n^n}{r^{\frac{n}{n+1}}} \tilde{\varepsilon} \left(\theta, \, n\right)$$

• Internal energy

$$U(\bar{\varepsilon}) = \frac{n\alpha (\sigma_p^0)^2}{E(n+1)} \left(\frac{E\bar{\varepsilon}}{\alpha \sigma_p^0}\right)^{\frac{n+1}{n}} \implies U \to \frac{n\alpha (\sigma_p^0)^2}{E(n+1)} \frac{k_n^{n+1}}{r} \widetilde{U}(\theta, n)$$

• J-integral

$$J = \lim_{r \to 0} \int_{-\pi}^{\pi} \left(U \boldsymbol{n}_{x} - \boldsymbol{u}_{,x} \cdot \boldsymbol{\sigma} \cdot \boldsymbol{n} \right) r d\theta = \frac{\alpha \left(\sigma_{p}^{0} \right)^{2}}{E} k_{n}^{n+1} \int_{-\pi}^{\pi} \tilde{h} \left(\theta, \, n \right) d\theta$$
$$\implies k_{n} = \left(\frac{JE}{\alpha \left(\sigma_{p}^{0} \right)^{2} I_{n}} \right)^{\frac{1}{n+1}} \text{ with } I_{n} = \int_{-\pi}^{\pi} \tilde{h} \left(\theta, \, n \right) d\theta$$



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- Solution in terms of J-integral
 - Near fields are obtained using $s' = -\frac{1}{n+1}$
 - Plastic stress intensity factor

$$k_n = \left(\frac{JE}{\alpha \left(\sigma_p^0\right)^2 I_n}\right)^{\frac{1}{n+1}}$$

• Stress field

$$\sigma(r, \theta) \to \sigma_p^0 k_n \frac{\tilde{\sigma}(\theta, n)}{r^{\frac{1}{n+1}}} \implies \sigma = \sigma_p^0 \left(\frac{JE}{r\alpha \left(\sigma_p^0\right)^2 I_n}\right)^{\frac{1}{n+1}} \tilde{\sigma}(\theta, n)$$
• Strain field
$$\varepsilon \to \frac{\alpha \sigma_p^0}{E} \frac{k_n^n}{r^{\frac{n}{n+1}}} \tilde{\varepsilon}(\theta, n) \implies \varepsilon = \frac{\sigma_p^0 \alpha}{E} \left(\frac{JE}{r\alpha \left(\sigma_p^0\right)^2 I_n}\right)^{\frac{n}{n+1}} \tilde{\varepsilon}(\theta, n)$$

• Limit case of $n \rightarrow 1$

$$J = \frac{K_I^2}{E'} \quad \Longrightarrow \quad \boldsymbol{\sigma} \propto \frac{K_I}{\sqrt{r}}$$



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- Resolution for a semi-infinite crack in plane ε state
 - Airy functions
 - Linear momentum $\boldsymbol{\sigma}_{\alpha\beta,\beta} + \boldsymbol{b}_{\alpha} = 0$
 - If **b** = 0, there exists an Airy function Φ : $\sigma_{\alpha\beta} = -\Phi_{,\alpha\beta} + \delta_{\alpha\beta}\Phi_{,\gamma\gamma}$
 - We are not in linear elasticity \Longrightarrow we cannot say $\nabla^2 \nabla^2 \Phi = 0$

$$egin{aligned} oldsymbol{\sigma}_{rr} &= rac{1}{r} \Phi_{,r} + rac{1}{r^2} \Phi_{, heta heta} \ oldsymbol{\sigma}_{ heta heta} &= \Phi_{,rr} \ oldsymbol{\sigma}_{r heta} &= -rac{1}{r} \Phi_{,r heta} + rac{1}{r^2} \Phi_{, heta} \end{aligned}$$

- In polar coordinates ٠
- $\left(\frac{(n+1)^{2}}{(n+1)^{2}}\right)$ We are seeking a solution in $r^{-1/(n+1)}$ \implies we choose Φ
- So the stress expressions satisfying the balance equation ٠

$$\sigma_{rr} = r^{\frac{-1}{n+1}} \left(\frac{n+1}{n} f + \frac{(n+1)^2}{n(2n+1)} f'' \right)$$

$$\sigma_{\theta\theta} = r^{\frac{-1}{n+1}} f$$

$$\sigma_{r\theta} = r^{\frac{-1}{n+1}} \left(-\frac{n+1}{n} + \frac{(n+1)^2}{n(2n+1)} \right) f' = r^{\frac{-1}{n+1}} \frac{-(n+1)}{(2n+1)} f'$$



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 $\frac{-}{n+1} + 2$

- Resolution for a semi-infinite crack in plane ε state (2)
 - Stream function
 - Since we are incompressible $\varepsilon_{\alpha\beta} = \frac{u_{\alpha,\beta} + u_{\beta,\alpha}}{2}$ & $\varepsilon_{\alpha\alpha} = u_{\alpha,\alpha} = 0$ with plane strain assumption (if not: $\varepsilon_{zz} \neq 0$)

• So displacements derive from a stream function Ψ with $\begin{cases} u_x = \Psi_{,y} \\ u_y = -\Psi_{,x} \end{cases}$

• In polar coordinates

-
$$\boldsymbol{u}_{x} = \Psi_{,y} = \sin\theta\Psi_{,r} + \frac{\cos\theta}{r}\Psi_{,\theta} \, \boldsymbol{\&} \, \boldsymbol{u}_{y} = -\Psi_{,x} = -\cos\theta\Psi_{,r} + \frac{\sin\theta}{r}\Psi_{,\theta}$$

> the displacements become
$$\begin{cases} \boldsymbol{u}_{r} = \cos\theta\boldsymbol{u}_{x} + \sin\theta\boldsymbol{u}_{y} = \frac{\Psi_{,\theta}}{r} \\ \boldsymbol{u}_{\theta} = -\sin\theta\boldsymbol{u}_{x} + \cos\theta\boldsymbol{u}_{y} = -\Psi_{,r} \end{cases}$$

Strains are obtained from

$$\begin{cases} \boldsymbol{\varepsilon}_{rr} = \boldsymbol{u}_{r,r} = \frac{\Psi_{,r\theta}}{r} - \frac{\Psi_{,\theta}}{r^2} \\ \boldsymbol{\varepsilon}_{\theta\theta} = \frac{\boldsymbol{u}_r}{r} + \frac{1}{r} \boldsymbol{u}_{\theta,\theta} = -\boldsymbol{\varepsilon}_{rr} \\ \boldsymbol{\varepsilon}_{r\theta} = \frac{1}{2} \left(\frac{\boldsymbol{u}_{r,\theta}}{r} + \boldsymbol{u}_{\theta,r} - \frac{\boldsymbol{u}_{\theta}}{r} \right) = \frac{1}{2} \left(\frac{\Psi_{,\theta\theta}}{r^2} - \Psi_{,rr} + \frac{\Psi_{,r}}{r} \right) \\ \text{seeking a solution in } r^{-n/(n+1)} \implies \text{we choose } \Psi = -r^{\frac{n+2}{n+1}} \boldsymbol{g}(\theta) \end{cases}$$



We are



- Resolution for a semi-infinite crack in plane ε state (3)
 - Strain tensor
 - Stream function $\Psi = -r^{\frac{n+2}{n+1}}g\left(\theta\right)$
 - So the strain components are

$$\begin{cases} \varepsilon_{rr} = \frac{\Psi_{,r\theta}}{r} - \frac{\Psi_{,\theta}}{r^2} = r^{\frac{-n}{n+1}} \left(-\frac{n+2}{n+1}g' + g' \right) = -\frac{1}{n+1}r^{\frac{-n}{n+1}}g' \\ \varepsilon_{\theta\theta} = \frac{1}{n+1}r^{\frac{-n}{n+1}}g' \\ \varepsilon_{r\theta} = \frac{1}{2} \left(\frac{\Psi_{,\theta\theta}}{r^2} - \Psi_{,rr} + \frac{\Psi_{,r}}{r} \right) = \frac{r^{\frac{-n}{n+1}}}{2} \left(-g'' + \frac{n+2}{(n+1)^2}g - \frac{n+2}{n+1}g \right) \\ = -\frac{r^{\frac{-n}{n+1}}}{2} \left(g'' + n\frac{n+2}{(n+1)^2}g \right) \end{cases}$$





- Resolution for a semi-infinite crack in plane ε state (4)
 - Stress & Strain tensors

$$\begin{cases} \boldsymbol{\sigma}_{rr} = r^{\frac{-1}{n+1}} \left(\frac{n+1}{n} f + \frac{(n+1)^2}{n(2n+1)} f'' \right) & \& \boldsymbol{\sigma}_{\theta\theta} = r^{\frac{-1}{n+1}} f \\ \boldsymbol{\sigma}_{r\theta} = r^{\frac{-1}{n+1}} \left(-\frac{n+1}{n} + \frac{(n+1)^2}{n(2n+1)} \right) f' = r^{\frac{-1}{n+1}} \frac{-(n+1)}{(2n+1)} f' \\ & \int \boldsymbol{\varepsilon}_{rr} = -\frac{1}{n+1} r^{\frac{-n}{n+1}} g' & \& \boldsymbol{\varepsilon}_{\theta\theta} = \frac{1}{n+1} r^{\frac{-n}{n+1}} g' \\ \boldsymbol{\varepsilon}_{r\theta} = -\frac{r^{\frac{-n}{n+1}}}{2} \left(g'' + n \frac{n+2}{(n+1)^2} g \right) \end{cases}$$

- 2 unknown functions f and $g \implies$ need for 2 equations

• The power law

• The normality equation

$$\bar{\varepsilon} = \frac{\alpha \sigma_p^0}{E} \left(\frac{\sigma_e}{\sigma_p^0}\right)^n$$
$$\varepsilon = \bar{\varepsilon} \sqrt{\frac{3}{2}} \frac{\mathbf{s}}{\sqrt{\mathbf{s}:\mathbf{s}}} = \frac{3\bar{\varepsilon}}{2\sigma_e} \mathbf{s}$$





• Resolution for a semi-infinite crack in plane ε state (5)

- Power law
$$\bar{\varepsilon} = \frac{\alpha \sigma_p^0}{E} \left(\frac{\sigma_e}{\sigma_p^0}\right)^n$$
 with normality $\varepsilon = \bar{\varepsilon} \sqrt{\frac{3}{2}} \frac{\mathbf{s}}{\sqrt{\mathbf{s}:\mathbf{s}}} = \frac{3\bar{\varepsilon}}{2\sigma_e} \mathbf{s}$

• These two equations lead to (see annex 1 for details)

$$\begin{split} & -\left[\frac{4}{3\left(1+n\right)^2}g'^2 + \frac{1}{3}\left(g''+n\frac{n+2}{\left(n+1\right)^2}g\right)^2\right] = \\ & \left(\frac{\alpha\sigma_p^0}{E\left(\sigma_p^0\right)^n}\right)^2 \left[\frac{3}{4}\left(\frac{1}{n}f + \frac{\left(n+1\right)^2}{n\left(2n+1\right)}f''\right)^2 + 3\left(\frac{n+1}{2n+1}\right)^2f'^2\right]^n \\ & - \frac{2}{2n+1}g'f' = -\frac{1}{2}\left[\frac{1}{n}f + \frac{\left(n+1\right)^2}{n\left(2n+1\right)}f''\right]\left[g''+n\frac{n+2}{\left(n+1\right)^2}g\right] \\ & \text{Back to HRR field} \end{split}$$

• Near the crack tip, the fields were normalized by introducing an intensity factor

$$\boldsymbol{\sigma}\left(r,\,\theta\right) \to \sigma_{p}^{0}k_{n}\frac{\tilde{\boldsymbol{\sigma}}\left(\theta,\,n\right)}{r^{\frac{1}{n+1}}} \quad \boldsymbol{\&} \ \boldsymbol{\varepsilon} \to \frac{\alpha\sigma_{p}^{0}}{E}\frac{k_{n}^{n}}{r^{\frac{n}{n+1}}}\tilde{\boldsymbol{\varepsilon}}\left(\theta,\,n\right)$$

• Therefore, the functions f and g are rewritten

$$f(\theta) = \sigma_p^0 k_n \tilde{f}(\theta) \quad \& \quad g(\theta) = \frac{\alpha \sigma_p^0}{E} k_n^n \tilde{g}(\theta)$$



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- Resolution for a semi-infinite crack in plane ε state (6)
 - Using the new non-dimensional functions leads to

$$\begin{cases} \left[\tilde{f}'' + \frac{2n+1}{(n+1)^2}\tilde{f}\right] \left[\tilde{g}'' + n\frac{n+2}{(n+1)^2}\tilde{g}\right] = -\frac{4n}{(n+1)^2}\tilde{g}'\tilde{f}' \\ \left[\frac{4}{(1+n)^2}\tilde{g}'^2 + \left(\tilde{g}'' + n\frac{n+2}{(n+1)^2}\tilde{g}\right)^2\right]^{\frac{1}{n}} = \\ \frac{3^{\frac{n+1}{n}}}{4}\frac{(n+1)^4}{n^2(2n+1)^2} \left[\left(\frac{2n+1}{(n+1)^2}\tilde{f} + \tilde{f}''\right)^2 + \frac{4n^2}{(n+1)^2}\tilde{f}'^2\right] \end{cases}$$

- These equations can be reduced to a differential equation of the 4th order in \tilde{f}

See appendix I

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$$\cdot \tilde{f}^{\prime\prime\prime\prime\prime} + \frac{2n+1}{\left(1+n\right)^{2}} \tilde{f}^{\prime\prime} + \frac{1}{n\tilde{l}^{2}+4\tilde{f}^{\prime2}} \left\{ \frac{n^{2}}{\left(1+n\right)^{2}} \left[\frac{4}{n} \tilde{f}^{\prime\prime} + \frac{n+2}{1+n} \tilde{l} \right] \left[\tilde{l}^{2}+4\tilde{f}^{\prime2} \right] + \frac{4n\left(n-1\right)}{\left(n+1\right)^{2}} \tilde{f}^{\prime} \left[\tilde{l}\tilde{l}^{\prime} + 4\tilde{f}^{\prime}\tilde{f}^{\prime\prime} \right] + \frac{n\left(n-1\right)}{n+1} \left[3\tilde{l}\tilde{l}^{\prime2} + 8\tilde{l}^{\prime}\tilde{f}^{\prime}\tilde{f}^{\prime\prime} + 4\tilde{l}\left(\tilde{f}^{\prime\prime2} + \tilde{f}^{\prime}\tilde{f}^{\prime\prime\prime} \right) \right] \\ + \frac{n\left(n-1\right)\left(n-3\right)}{n+1} \frac{\tilde{l}}{\tilde{l}^{2}+4\tilde{f}^{\prime2}} \left[\tilde{l}\tilde{l}^{\prime} + 4\tilde{f}^{\prime}\tilde{f}^{\prime\prime\prime} \right]^{2} \right\} \& \tilde{l} = \frac{1}{n} \left[\left(n+1\right)\tilde{f}^{\prime\prime} + \frac{2n+1}{n+1}\tilde{f} \right]$$





- Resolution for a semi-infinite crack in plane ε state (7)
 - New differential equation of the 4th order in \tilde{f}
 - Stress field from \tilde{f}

$$\left(\boldsymbol{\sigma}_{rr} = r^{\frac{-1}{n+1}} \left(\frac{n+1}{n} f + \frac{(n+1)^2}{n(2n+1)} f'' \right) & \mathbf{\delta} \quad \boldsymbol{\sigma}_{\theta\theta} = r^{\frac{-1}{n+1}} f \\ \boldsymbol{\sigma}_{r\theta} = r^{\frac{-1}{n+1}} \left(-\frac{n+1}{n} + \frac{(n+1)^2}{n(2n+1)} \right) f' = r^{\frac{-1}{n+1}} \frac{-(n+1)}{(2n+1)} f'$$

- Range-Kutta resolution of differential equation of the 4th order in \tilde{f}

- Initial boundary conditions
 - Mode I symmetry: $\boldsymbol{\sigma}_{r\theta} (\theta = 0) = \boldsymbol{\sigma}_{rr,\theta} (\theta = 0) = \boldsymbol{\sigma}_{\theta\theta,\theta} (\theta = 0) = 0$ $\implies \tilde{f}'(0) = \tilde{f}'''(0) = 0$

- Stress free lips: $\boldsymbol{\sigma}_{r\theta} \left(\theta = \pi \right) = \boldsymbol{\sigma}_{\theta\theta} \left(\theta = \pi \right) = 0 \Longrightarrow \tilde{f} \left(\pi \right) = \tilde{f}' \left(\pi \right) = 0$

 \implies iterations on $\tilde{f}''(0)$ until these 2 conditions are satisfied

- It remains to choose $\,\, \widetilde{f}\left(0
 ight)$
- The differential equation is valid for any multiple of
 - Initial condition $\tilde{f}(0)$ is chosen so that $\max_{\theta} \tilde{\sigma}_{e}(\theta, n) = 1$





- Resolution for a semi-infinite crack in plane ε state (8)
 - Solution of the plane ϵ problem



•
$$\boldsymbol{\sigma} = \sigma_p^0 \left(\frac{JE}{r\alpha \left(\sigma_p^0 \right)^2 I_n} \right)^{n+1} \tilde{\boldsymbol{\sigma}} \left(\theta, n \right) \& \quad \boldsymbol{\varepsilon} = \frac{\sigma_p^0 \alpha}{E} \left(\frac{JE}{r\alpha \left(\sigma_p^0 \right)^2 I_n} \right)^{n+1} \tilde{\boldsymbol{\varepsilon}} \left(\theta, n \right)$$

- J characterizes the loading intensity on the crack
 - Problem dependent
- I_n is a normalizing factor allowing to express the dependency with respect to J
 - Still needs to be evaluated







• Determination of stress and strain fields



• Process zone shape

۲

The elastic-plastic boundary corresponds to $\sigma_e = \sigma_p^0$

Since
$$\boldsymbol{\sigma} = \sigma_p^0 \left(\frac{JE}{r\alpha \left(\sigma_p^0\right)^2 I_n} \right)^{\frac{1}{n+1}} \tilde{\boldsymbol{\sigma}} \left(\theta, n\right)$$

 $\implies 1 = \frac{\sigma_e}{\sigma_p^0} = \left(\frac{JE}{\alpha \left(\sigma_p^0\right)^2} \right)^{\frac{1}{n+1}} r^{-\frac{1}{n+1}} \tilde{\sigma}_e I_n^{-\frac{1}{n+1}}$

• Definition of the non-dimentional radius $\tilde{r} = \frac{r\alpha \left(\sigma_p^0\right)^2}{JE}$, $\implies 1 = \frac{\sigma_e}{\sigma_p^0} = \left(\frac{JE}{\alpha \left(\sigma_p^0\right)^2}\right)^{\frac{1}{n+1}} r^{-\frac{1}{n+1}} \tilde{\sigma}_e I_n^{-\frac{1}{n+1}}$ $\tilde{r}^{-\frac{1}{n+1}}$

 \implies the boundary is obtained for $\tilde{r}(\theta, n) = \frac{\left(\tilde{\sigma}_e(\theta, n)\right)^{n+1}}{I_n}$





Process zone shape (2) - The elastic-plastic boundary corresponds to $\sigma_e = \sigma_p^0$ • In terms of $\tilde{r} = \frac{r \alpha \left(\sigma_p^0\right)^2}{JE}$, the boundary is obtained for $\tilde{r}(\theta, n) = \frac{(\tilde{\sigma}_e(\theta, n))^{n+1}}{I_n}$ n = 1.1 n = 1.1 0.2 n = 3 n = 3n = 13 n = 130.1 12 0 *i*~ 0 -0.1 -1 -0.2 -2 0.2 -0.2 -0.1 0.1 0 -5 -10 5 0 x 10⁻³ \tilde{x} x 10⁻³ х • Limit case of $n \to 1$: $J = \frac{K_I^2}{E} (1 - \nu^2) \implies r \propto (1 - \nu^2) \left(\frac{K_I}{\sigma_n^0}\right)^2$ 2021-2022 Fracture Mechanics – NLFM – HRR Theory 31



- The method does not remove the stress singularity (except for $n \rightarrow \infty$)
- Since the abscissa is $\tilde{r} = \frac{r\alpha (\sigma_p^0)^2}{JE}$, it appears that *J* is a measure of the intensity of the singular crack tip field (except for $n \rightarrow \infty$)







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- Crack tip opening displacement (CTOD) (3)
 - The CTOD is defined as $\delta(r)$ such that a 90°-angle is intercepted (3)

•
$$\delta_t = 2\boldsymbol{u}_y(r^*, \pi) = 2\left(\frac{\alpha\sigma_p^0}{E}\right)^{\frac{1}{n}} \left(\frac{J}{\sigma_p^0 I_n}\right)^{\frac{1}{n}}$$

 $[\tilde{\boldsymbol{u}}_x(\pi, n) + \tilde{\boldsymbol{u}}_y(\pi, n)]^{\frac{1}{n}} \tilde{\boldsymbol{u}}_y(\pi, n)$

$$\implies \delta_t = d_n \frac{J}{\sigma_p^0}$$

• d_n depends on

-
$$n$$

- but also on $\frac{\alpha \sigma_p^0}{E}$

 Since for a given material *J* is uniquely related to the CTOD, this last one can be used as a crack initiation criterion



 \bar{u}_x

 r^*

 δ_t

 u_{v}



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y

 \boldsymbol{X}



Summary

- Assumptions
 - J2-plasticity with power law description
 - Small deformations
 - There is no unloading and loading is proportional in all the directions (ok for crack initiation and not for crack propagation)
 - Elastic strains are assimilated to plastic strain (material is **incompressible**)
 - Semi-infinite crack
- HRR results for semi-infinite mode I crack

in plane ε state

Asymptotic stress, strain and displacement fields

$$\boldsymbol{\sigma} = \sigma_p^0 \left(\frac{JE}{r\alpha \left(\sigma_p^0 \right)^2 I_n} \right)^{\frac{1}{n+1}} \tilde{\boldsymbol{\sigma}} \left(\theta, n \right)$$

- The *J*-integral plays the role of an equivalent "plastic strain intensity factor"
- Stress field evolves in a proportional way, so this is applicable to incremental plasticity as long as *J* increases







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• Mode I crack in plane σ state

- Analysis is the same with other $\tilde{\sigma}(\theta, n)$, $\tilde{\varepsilon}(\theta, n)$ & $\tilde{u}(\theta, n)$ fields





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Other HRR solutions



So for the same *J*, a thin specimen is less stressed at crack tip







Other HRR solutions





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Other HRR solutions

- Mode I crack in plane σ state (4)
 - Process zone shapes
 - The process zones
 - Have a different shape
 - Are more diffuse in plane $\boldsymbol{\sigma}$



- In SSY, the process zone has a circular shape
 - Perfectly plastic





•



- Mode II and mixed mode crack
 - In SSY, the solution depends on the elastic mixity parameter

$$M^e = \frac{2}{\pi} \arctan \frac{K_I}{K_{II}}$$

- Examples
 - Process zones in plane ϵ state
 - SSY

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Compressibility

- Elastic part of the deformations in not incompressible
- Considering this effect will diffuse the plastic zone
 - Example: Mode I, plane ϵ state & SSY
 - The plastic zone size r_p is defined as the length of the plastic zone ahead of the crack











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- HRR solution in SSY can explain 3D effect (2)
 - During a toughness test, the *K* measured is an average one
 - It is more important for thin specimen
 - The process is never really plane $\boldsymbol{\epsilon}$
 - There are actually complex 3D effects
 - In SSY, even for a thin specimen, near the mid-plane a plane-ε-state is developed
 - If the load increases, the plastic zone can be plane-σ–like near the mid-plane
 - Effect of *T*-stress should also be considered
 - Recall *T*-stress is the 0-order term obtained with the asymptotic solution, which is dominant at radius r_c
 - In general, if the test is such that T < 0, the measured fracture K will be larger than for T > 0, independently of the thickness *
 - For ASTM toughness tests, the thickness

is large
$$t > 2.5 \left(\frac{K_C}{\sigma_p^0}\right)^2$$
 so that $T > 0$





*D.J Smith, M.R Ayatollahi and M.J Pavier, *Proc. R. Soc. A, 2006, vol.* 462, pp 415-2437





- Effective crack length for SSY
 - If SSY assumption holds
 - J can be expressed in terms of K
 - Then the plastic size can be written $r \propto rac{JE}{\left(\sigma_n^0
 ight)^2} \propto$
 - However, there are dependencies on
 - Parameters n & v
 - Whether it is plane ϵ or plane σ
 - Rice's model for perfectly plastic material
 - Intersection of the linear elastic solution with the yield stress leads at ηr_p

$$\frac{K_I}{\sqrt{2\pi\eta r_p}} = \sigma_p^0 \implies \sqrt{\eta r_p} = \frac{1}{\sqrt{2\pi}} \frac{K_I}{\sigma_p^0}$$

- But there is a redistribution of the stresses so that the traction remains the same
 - First order approximation: stress distribution is shifted

$$\int_{0}^{\eta r_{p}} \left(\boldsymbol{\sigma}_{yy} - \boldsymbol{\sigma}_{p}^{0} \right) dr = \boldsymbol{\sigma}_{p}^{0} \left(1 - \eta \right) r_{p}$$





J





- Effective crack length for SSY (3)
 - Rice's model (2)
 - So everything is as if the crack had

an effective length $a + \eta r_p = a + r_p/2$

$$\begin{cases} r_p = \frac{1}{\pi} \left(\frac{K_I}{\sigma_p^0} \right)^2 \\ \eta = \frac{1}{2} \end{cases}$$



- From HRR models, numerical simulations, etc ...

• Considering blunting, compressibility, hardening, ..., an **estimation** is

$$\eta r_p = \frac{r_p}{2} = \begin{cases} \frac{n-1}{n+1} \frac{1}{2\pi} \left(\frac{K_I(a_{\text{eff}})}{\sigma_p^0}\right)^2 & \text{if plane } \sigma \\ \frac{n-1}{n+1} \frac{1}{6\pi} \left(\frac{K_I(a_{\text{eff}})}{\sigma_p^0}\right)^2 & \text{if plane } \varepsilon \end{cases}$$







- Effective crack length for SSY (4)
 - If σ_{∞} < 50% of σ_{p}^{0} then a second order SSY assumption holds
 - The cohesive zone remains small compared to crack size
 - The effective crack size can be stated as $a + \eta r_p = a + r_p/2$

with
$$\eta r_p = \frac{r_p}{2} = \begin{cases} \frac{n-1}{n+1} \frac{1}{2\pi} \left(\frac{K_I(a_{\text{eff}})}{\sigma_p^0} \right)^2 & \text{if plane } \sigma \\ \frac{n-1}{n+1} \frac{1}{6\pi} \left(\frac{K_I(a_{\text{eff}})}{\sigma_p^0} \right)^2 & \text{if plane } \varepsilon \end{cases}$$
 for cracks in finite plates

- So there is an iterative procedure to follow:
 - a) compute *K* from *a*
 - b) compute effective crack size
 - c) compute new K from a_{eff} and back to b) if needed
- This method is a correction for linear fracture mechanics, but does not allow considering problems with large yielding





Validity in SSY

- We have two asymptotic solutions
 - HRR field is valid in the process zone •
 - LEFM is still valid in the elastic zone close to the crack tip •



Conditions

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This is the case if all sizes are 25 times larger than the plastic zone •

$$a, t, L > 25r_p \simeq \frac{25}{3\pi} \left(\frac{K_I}{\sigma_p^0}\right)^2 \simeq 2.5 \left(\frac{K_I}{\sigma_p^0}\right)^2$$







- Validity in SSY (2)
 - Crack initiation criteria
 - Criteria based on *J* or δ_t are valid: $J \ge J_C$ or $\delta_t \ge \delta_C$
 - $J \& \delta_t$ depend on a, the geometry, the loading, ...
 - But as the LEFM solution holds, we can still use $K(a) \ge K_C$
 - Might be corrected by using the effective length a_{eff} if σ_{∞} < 50% of σ_{p}^{0}
 - Sizes for K-based toughness test

$$a, t, L > 2.5 \left(\frac{K_C}{\sigma_p^0}\right)^2$$



- Examples:
 - Titanium alloy 6%AI-4%V
 - » Yield: 830 MPa

 - » a, t, L > 1.1 cm
 - Strength steel
 - » Yield: 350 MPa
 - » Toughness: 250 MPa \cdot $m^{{}^{1\!\!/_2}}$
 - » *a, t, L* > 1.27 m !!!





- Validity in elasto-plastic conditions
 - Deformations are small
 - We still have one asymptotic solution valid
 - HRR field is valid in the process zone
 - LEFM is **NOT** valid in the elastic zone close to the crack



- Conditions
 - This is the case if all sizes are 25 times larger than CTOD

$$a, t, L > 25\delta_t \simeq 25 \frac{J}{\sigma_p^0}$$







- Validity in elasto-plastic conditions (2)
 - Crack initiation criteria
 - Criteria based on *J* or δ_t are valid: $J \ge J_C$ or $\delta_t \ge \delta_C$
 - *J* & δ_t depend on *a*, the geometry, the loading, ...
 - The LEFM solution **DOES NOT** hold, we **CANNOT** use $K(a) \ge K_C$
 - Sizes for J-based toughness test

$$a, t, L > 25 \frac{J_C}{\sigma^0}$$

- If K_c is computed from J_c : $a, t, L > 25 \frac{K_C^2 (1 \nu^2)}{E\sigma_n^0}$
- Examples:
 - Titanium alloy 6%AI-4%V
 - » Yield: 830 MPa
 - » Toughness: 55 MPa \cdot $m^{{}^{1\!\!/_2}}$
 - » Young: 110 GPa
 - » a, t, L > 0.75 mm
 - Strength steel
 - » Yield: 350 MPa
 - » Toughness: 250 MPa \cdot $m^{{}^{1\!\!/_2}}$
 - » Young: 210 GPa
 - » a, t, L > 1.93 cm







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- Validity in large yielding
 - Example: ligament size is too small
 - Small deformations assumption does not hold _
 - Neither HRR field nor LEFM asymptotic fields are valid _



Crack initiation criterion?

• As there is no zone of *J*-dominance can *J* still be used?







- Validity in large yielding (2)
 - Plastic strain concentrations depend on the experiment
 - Zones near free boundaries or other cracks tend to be less stressed



– Crack initiation criterion?

- Solution is no longer uniquely governed by J
- Relation between $J \& \delta_t$ is dependent on the configuration and on the loading
- The critical J_c measured for an experiment might not be valid for another one
- A 2-parameter characterization is needed





Exercise 1: Specimen with centered crack

- Thick specimen with centered crack:
 - Steel
 - Yield: 350 MPa
 - Toughness: 250 MPa \cdot m^{1/2} •
 - Young: 210 GPa
 - Hardening exponent >>
 - Loading
 - *P* = 25, 50 & 150 kN
 - Compute the stress intensity factor
 - in terms of crack size a
 - Compare the solutions obtained by
 - LEFM
 - LEFM with effective crack length
 - Check validity









• *P*= 50 kN

- Far stress field

•
$$\sigma_{\infty} = \frac{P}{2Wt} = \frac{50000}{0.038\ 0.0127} = 103.6 \text{ MPa}$$

- Solution for a = 7 mm

• SIF:
$$K_I = \sigma_{\infty} \sqrt{\pi a} f\left(\frac{a}{W}\right)$$

with $f\left(\frac{a}{W}\right) = \left[1 - 0.025 \left(\frac{a}{W}\right)^2 + 0.06 \left(\frac{a}{W}\right)^4\right] \sqrt{\frac{1}{\cos \frac{\pi a}{2W}}}$
 $\implies K_I = 103.6 \, 10^6 \sqrt{\pi 0.007} f(0.3684) = 16.75 \text{ MPa} \sqrt{\text{m}}$





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- *P*= 50 kN (2)
 - Solution for a = 7 mm (2)
 - Effective crack length
 - First iteration

Second iteration

$$\begin{cases} r_p^{(1)} = \frac{1}{3\pi} \left(\frac{K_I}{\sigma_p^0}\right)^2 = \frac{1}{3\pi} \left(\frac{16.75}{350}\right)^2 = 0.24 \text{ mm} \\ a_{\text{eff}}^{(1)} = a + \frac{r_p}{2} = 7.12 \text{ mm} \\ K_I^{(1)} = 103.6 \, 10^6 \sqrt{\pi 0.00712} f(0.3747) = 16.95 \text{ MPa} \sqrt{\text{m}} \\ \begin{cases} r_p^{(2)} = \frac{1}{3\pi} \left(\frac{K_I^{(1)}}{\sigma_p^0}\right)^2 = \frac{1}{3\pi} \left(\frac{16.95}{350}\right)^2 = 0.249 \text{ mm} \\ a_{\text{eff}}^{(2)} = a + \frac{r_p}{2} = 7.1245 \text{ mm} \\ K_I^{(2)} = 103.6 \, 10^6 \sqrt{\pi 0.0071245} f(0.3750) = 16.96 \text{ MPa} \sqrt{\text{m}} \end{cases}$$

- It has converged
- The correction is of about 1%
- Validity:
 - σ_∞< 0.5 σ_p⁰: OK
 - $a = 7 \text{ mm}, t = 12.7 \text{ mm}, \& W-a = 12 \text{ mm} > 25 r_p = 6.2 \text{ mm}$: OK





• *P*= 50 kN (3)



 $-r_p = 0.383 \text{ mm} \implies 25 r_p = 9.6 \text{ mm} > a$: solution not valid anymore





• *P*= 25 kN



anymore





P = 150 kN



 $-25 r_p > a$ always

- As σ_{∞} = 0.88 σ_{p}^{0} SSY theory does not hold, the size of the plastic zone cannot be approximated by $r_p = \frac{1}{3\pi} \left(\frac{K_I}{\sigma_r^0} \right)^2$

We will use the HRR theory, but we need to evaluate the J-integral _





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• Resolution for a semi-infinite crack in plane ε state (A1)

- Power law
$$\bar{\varepsilon} = \frac{\alpha \sigma_p^0}{E} \left(\frac{\sigma_e}{\sigma_p^0}\right)^n$$
 with $\sigma_e = \sqrt{\frac{3}{2}\mathbf{s} : \mathbf{s}}$

- Deviatoric tensor, plane ϵ & incompressible material:

$$-\operatorname{As} \varepsilon = \overline{\varepsilon} \sqrt{\frac{3}{2}} \frac{\mathbf{s}}{\sqrt{\mathbf{s}:\mathbf{s}}} = \frac{3\overline{\varepsilon}}{2\sigma_e} \mathbf{s} \Longrightarrow \mathbf{s}_{zz} = 0 \text{ or } \sigma_{zz} - \frac{\sigma_{rr} + \sigma_{\theta\theta} + \sigma_{zz}}{3} = 0$$

$$-\operatorname{Out} \text{ of plane stress } \sigma_{zz} = \frac{\sigma_{rr} + \sigma_{\theta\theta}}{2} \mathbf{s} \operatorname{tr} \sigma = \frac{3}{2} \left(\sigma_{rr} + \sigma_{\theta\theta} \right)$$

$$-\operatorname{Non zero components of deviatoric tensor: } \mathbf{s} = \left(\begin{array}{c} \frac{\sigma_{rr} - \sigma_{\theta\theta}}{2} & \sigma_{r\theta} \\ \sigma_{r\theta} & \frac{\sigma_{\theta\theta} - \sigma_{rr}}{2} \end{array} \right)$$

$$\cdot \text{ Equivalent von Mises stress } \sigma_e = \sqrt{\frac{3}{2}} \sqrt{\mathbf{s}:\mathbf{s}} = \sqrt{\frac{3}{4}} \left(\sigma_{rr} - \sigma_{\theta\theta} \right)^2 + 3\sigma_{r\theta}^2}$$

- In terms of r and f

$$\implies \sigma_e^2 = r^{\frac{-2}{n+1}} \left[\frac{3}{4} \left(\frac{1}{n} f + \frac{\left(n+1\right)^2}{n\left(2n+1\right)} f'' \right)^2 + 3 \left(\frac{n+1}{2n+1} \right)^2 f'^2 \right]$$





- Resolution for a semi-infinite crack in plane ε state (A2)
 - Power law (2)
 - Equivalent deformation in terms of r and g

$$\begin{split} \bar{\varepsilon}^2 &= \frac{2}{3} \left(\varepsilon_{rr}^2 + \varepsilon_{\theta\theta}^2 + 2\varepsilon_{r\theta}^2 \right) = \frac{2}{3} r^{\frac{-2n}{n+1}} \left[\frac{2}{(1+n)^2} g'^2 + \frac{1}{2} \left(g'' + n \frac{n+2}{(n+1)^2} g \right)^2 \right] \\ \bullet \text{ So the power law } \bar{\varepsilon}^2 &= \left(\frac{\alpha \sigma_p^0}{E} \right)^2 \left(\frac{\sigma_e}{\sigma_p^0} \right)^{2n} \text{ becomes} \\ \left[\frac{4}{3 \left(1+n \right)^2} g'^2 + \frac{1}{3} \left(g'' + n \frac{n+2}{(n+1)^2} g \right)^2 \right] = \\ \left(\frac{\alpha \sigma_p^0}{E \left(\sigma_p^0 \right)^n} \right)^2 \left[\frac{3}{4} \left(\frac{1}{n} f + \frac{(n+1)^2}{n \left(2n+1 \right)} f'' \right)^2 + 3 \left(\frac{n+1}{2n+1} \right)^2 f'^2 \right]^n \end{split}$$

- We have a differential equation in terms of functions depending on θ & n
- But we are still missing a relation between the functions f & g

We will now study the normality relation $\varepsilon = \overline{\varepsilon} \sqrt{\frac{3}{2}} \frac{\mathbf{s}}{\sqrt{\mathbf{s}:\mathbf{s}}} = \frac{3\overline{\varepsilon}}{2\sigma_e} \mathbf{s}$





Annex 1: Mode I crack and HRR theory

Resolution for a semi-infinite crack in plane ε state (A3) ۲

- Normality
$$\boldsymbol{\varepsilon} = \bar{\varepsilon} \sqrt{\frac{3}{2}} \frac{\mathbf{s}}{\sqrt{\mathbf{s} : \mathbf{s}}} = \frac{3\bar{\varepsilon}}{2\sigma_e} \mathbf{s} = \frac{3\bar{\varepsilon}}{2\sigma_e} \begin{pmatrix} \frac{\boldsymbol{\sigma}_{rr} - \boldsymbol{\sigma}_{\theta\theta}}{2} & \boldsymbol{\sigma}_{r\theta} \\ \boldsymbol{\sigma}_{r\theta} & \frac{\boldsymbol{\sigma}_{\theta\theta} - \boldsymbol{\sigma}_{rr}}{2} \end{pmatrix}$$

• Strain-stress relations $\varepsilon_{r\theta} = \frac{3\bar{\varepsilon}}{2\sigma_e} \sigma_{r\theta} \& \varepsilon_{rr} - \varepsilon_{\theta\theta} = \frac{3\bar{\varepsilon}}{2\sigma_e} (\sigma_{rr} - \sigma_{\theta\theta})$

imply $(\boldsymbol{\varepsilon}_{rr} - \boldsymbol{\varepsilon}_{\theta\theta}) \boldsymbol{\sigma}_{r\theta} = (\boldsymbol{\sigma}_{rr} - \boldsymbol{\sigma}_{\theta\theta}) \boldsymbol{\varepsilon}_{r\theta}$

• In terms of
$$f \& g$$
: $\frac{2}{2n+1}g'f' = -\frac{1}{2}\left[\frac{1}{n}f + \frac{(n+1)^2}{n(2n+1)}f''\right]\left[g'' + n\frac{n+2}{(n+1)^2}g\right]$

- Back to HRR field
 - Near the crack tip, the fields were normalized by introducing an intensity factor

$$\boldsymbol{\sigma}\left(r,\,\theta\right) \to \sigma_{p}^{0}k_{n}\frac{\tilde{\boldsymbol{\sigma}}\left(\theta,\,n\right)}{r^{\frac{1}{n+1}}} \quad \& \quad \boldsymbol{\varepsilon} \to \frac{\alpha\sigma_{p}^{0}}{E}\frac{k_{n}^{n}}{r^{\frac{n}{n+1}}}\tilde{\boldsymbol{\varepsilon}}\left(\theta,\,n\right)$$

• Therefore, the functions *f* and *g* are rewritten

$$f(\theta) = \sigma_p^0 k_n \tilde{f}(\theta) \quad \& \quad g(\theta) = \frac{\alpha \sigma_p^0}{E} k_n^n \tilde{g}(\theta)$$

• Which allows writing the differential equations in terms of \tilde{f} & \tilde{g}







• The two differential equations are

$$-\left[\tilde{f}'' + \frac{2n+1}{(n+1)^2}\tilde{f}\right] \left[\tilde{g}'' + n\frac{n+2}{(n+1)^2}\tilde{g}\right] = -\frac{4n}{(n+1)^2}\tilde{g}'\tilde{f}' \qquad \&$$

$$-\left[\frac{4}{(1+n)^2}\tilde{g}'^2 + \left(\tilde{g}'' + n\frac{n+2}{(n+1)^2}\tilde{g}\right)^2\right]^{\frac{1}{n}} =$$

$$\frac{3^{\frac{n+1}{n}}}{4}\frac{(n+1)^4}{n^2(2n+1)^2} \left[\left(\frac{2n+1}{(n+1)^2}\tilde{f} + \tilde{f}''\right)^2 + \frac{4n^2}{(n+1)^2}\tilde{f}'^2\right]$$

$$-\operatorname{Let}\tilde{l} = \frac{1}{n}\left[(n+1)\tilde{f}'' + \frac{2n+1}{n+1}\tilde{f}\right] \text{ then, these equations become}$$

$$\cdot \left[\frac{4}{(1+n)^2}\tilde{g}'^2 + \left(\tilde{g}'' + n\frac{n+2}{(n+1)^2}\tilde{g}\right)^2\right]^{\frac{1}{n}} = \frac{3^{\frac{n+1}{n}}}{4}\frac{(n+1)^2}{(2n+1)^2}\left[\tilde{l}^2 + 4\tilde{f}'^2\right]$$

$$\cdot \tilde{l}\left[\tilde{g}'' + n\frac{n+2}{(n+1)^2}\tilde{g}\right] = -\frac{4}{(n+1)}\tilde{g}'\tilde{f}'$$



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• Theses equations can be reduced to a single ODE as

$$-\left[\frac{4}{(1+n)^2} + \frac{16}{(n+1)^2 \tilde{l}^2} \tilde{f}'^2\right]^{\frac{1}{n}} \tilde{g}'^{\frac{2}{n}} = \frac{3^{\frac{n+1}{n}}}{4} \frac{(n+1)^2}{(2n+1)^2} \left[\tilde{l}^2 + 4\tilde{f}'^2\right]$$
$$\Longrightarrow \tilde{g}' = \sqrt{\left[\frac{\frac{3^{n+1}}{4^n} \frac{(n+1)^{2n}}{(2n+1)^{2n}} \left[\tilde{l}^2 + 4\tilde{f}'^2\right]^n}{\left[\frac{4}{(1+n)^2} + \frac{16}{(n+1)^2 \tilde{l}^2} \tilde{f}'^2\right]}\right]}$$
$$- \text{ After differentiating } \tilde{l} \left[\tilde{g}'' + n\frac{n+2}{(n+1)^2}\tilde{g}\right] = -\frac{4}{(n+1)}\tilde{g}'\tilde{f}' \text{ and }$$

substituting \tilde{g}' and its derivatives, it yields the fourth order ODE

$$\begin{split} \tilde{f}^{\prime\prime\prime\prime\prime} &+ \frac{2n+1}{\left(1+n\right)^{2}} \tilde{f}^{\prime\prime} + \frac{1}{n\tilde{l}^{2}+4\tilde{f}^{\prime2}} \left\{ \frac{n^{2}}{\left(1+n\right)^{2}} \left[\frac{4}{n} \tilde{f}^{\prime\prime} + \frac{n+2}{1+n} \tilde{l} \right] \left[\tilde{l}^{2}+4\tilde{f}^{\prime2} \right] + \\ \frac{4n\left(n-1\right)}{\left(n+1\right)^{2}} \tilde{f}^{\prime} \left[\tilde{l}\tilde{l}^{\prime} + 4\tilde{f}^{\prime}\tilde{f}^{\prime\prime} \right] + \frac{n\left(n-1\right)}{n+1} \left[3\tilde{l}\tilde{l}^{\prime2} + 8\tilde{l}^{\prime}\tilde{f}^{\prime}\tilde{f}^{\prime\prime} + 4\tilde{l}\left(\tilde{f}^{\prime\prime2} + \tilde{f}^{\prime}\tilde{f}^{\prime\prime\prime} \right) \right] \\ &+ \frac{n\left(n-1\right)\left(n-3\right)}{n+1} \frac{\tilde{l}}{\tilde{l}^{2}+4\tilde{f}^{\prime2}} \left[\tilde{l}\tilde{l}^{\prime} + 4\tilde{f}^{\prime}\tilde{f}^{\prime\prime\prime} \right]^{2} \right\} \end{split}$$



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Annex 2: Slip line solution for perfectly plastic material

- Semi-infinite mode I crack in plane ε state
 - Von Mises stress $\sigma_e = \sqrt{\frac{3}{2}}\sqrt{\mathbf{s} \cdot \mathbf{s}} = \sqrt{\frac{3}{4}}\left(\boldsymbol{\sigma}_{rr} \boldsymbol{\sigma}_{\theta\theta}\right)^2 + 3\boldsymbol{\sigma}_{r\theta}^2$ (Plane ε)
 - Perfectly plastic material
 - Von Mises stress: $\left(\frac{\sigma_p^0}{\sqrt{3}}\right)^3 = \left(\frac{\sigma_{rr} \sigma_{\theta\theta}}{2}\right)^2 + \sigma_{r\theta}^2$ The Mohr's circle is centered on $\frac{\sigma_{rr} + \sigma_{\theta\theta}}{2}$ with radius $\frac{\sigma_p^0}{\sqrt{3}}$
 - In Region *II* $\sigma_{rr} = \sigma_{\theta\theta}$ in frame
 - e_r , e_{θ} , we have pure shearing in the directions 0 and $\varphi = \pi/2$
 - e_x , e_y , we have pure shearing in the directions θ and $\varphi=\theta+\pi/2$





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Annex 2: Slip line solution for perfectly plastic material

- Semi-infinite mode I crack in plane ε state (2)
 - In Region *I* for $\theta \to 0$: $\sigma_{r\theta} \to 0$ \Longrightarrow in frame
 - e_r , e_{θ} , we have pure shearing in the directions $\varphi = \pi/4$ and $\varphi = -\pi/4$
 - e_x , e_y , we have pure shearing in the directions $\varphi = \pi/4$ and $\varphi = -\pi/4$
 - In Region *I* for $\theta \to \pi$: $\sigma_{r\theta} \to 0 \implies$ in frame
 - e_r , e_{θ} , we have pure shearing in the directions $\varphi = \pi/4$ and $\varphi = -\pi/4$
 - e_x , e_y , we have pure shearing in the directions $\varphi = -3\pi/4$ and $\varphi = 3\pi/4$
 - Slip directions are the ones of maximal shearing











- Mode I crack in plane σ state
 - Slip lines (perfectly plastic material)





